



# A strongly cooperative spinel nanohybrid as an efficient bifunctional oxygen electrocatalyst for oxygen reduction reaction and oxygen evolution reaction

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## ARTICLE INFO

### Keywords:

Bifunctional oxygen electrocatalyst  
Spinel nanohybrid  
Zn-air batteries  
Ultrathin nanosheets

## ABSTRACT

The development of efficient, stable and low-cost bifunctional oxygen electrocatalysts is critical to the realization of practically viable rechargeable Zn-air batteries. Herein, we report a strongly cooperative spinel nanohybrid as a promising air electrode catalyst for rechargeable Zn-air batteries. Ultrafine sub-10 nm  $\text{MnFe}_2\text{O}_4$  crystals are *in situ* grown on the ultrathin  $\text{NiCo}_2\text{O}_4$  nanosheets, leading to a highly effective surface area and a strong synergistic chemical coupling effect. The distinct architecture and complex composition endow an excellent bifunctional oxygen electrocatalytic activity in alkaline condition. The practical rechargeable Zn-air battery with the hybrid electrocatalyst demonstrates a high round-trip efficiency (a low discharge-charge voltage gap of 0.81 V at a reversible current density of  $10 \text{ mA cm}^{-2}$ ) and an outstanding durability, which outperforms the commercial Pt/Ru/C electrocatalyst. The resulting hybrid ( $\text{MnFe}_2\text{O}_4/\text{NiCo}_2\text{O}_4$ ) shows great promise as an alternative bifunctional electrocatalyst to the precious metals for the application in Zn-air batteries.

## 1. Introduction

As a promising post lithium-ion technology, rechargeable Zn-air batteries have attracted intense attention in the past decades due to their high theoretical energy and power density [1,2]. However, the sluggish kinetics of the oxygen reduction/evolution reaction (ORR/OER) has been the technical hurdle for the Zn-air battery application [3]. Thus, the discovery of bifunctional electrocatalysts which could efficiently drive both ORR and OER is of paramount importance. Presently, the state-of-art bifunctional electrocatalysts are noble-metal based catalysts, such as Pt/RuO<sub>2</sub> or Pt/IrO<sub>2</sub>. However, the commercialization of Zn-air battery is greatly hampered by two fundamental factors associated to the Pt/RuO<sub>2</sub> or Pt/IrO<sub>2</sub>: (1) the high cost, scarcity of the noble metal, and (2) the poor stability of the electrocatalysts in cyclic charging/discharging environment [4–6]. Therefore, it is highly desirable yet challenging to design non-noble-metal-based bifunctional electrocatalysts with efficient catalytic activity and good stability.

Mixed-valence transition metal oxides have emerged as an alternative oxygen electrocatalyst, among which a two-dimensional  $\text{NiCo}_2\text{O}_4$  nanosheets (NSs) electrocatalysts with an ultrathin feature has drawn particular interest owing to its excellent intrinsic OER activity, unique architecture, rich electroactive sites, and good electronic/ionic conductivity [7–12]. However, an insufficient ORR activity was found on  $\text{NiCo}_2\text{O}_4$  electrocatalysts, which greatly limits their application in

rechargeable Zn-air batteries [13,14]. Therefore, considerable efforts are required to advance the bifunctionality of the  $\text{NiCo}_2\text{O}_4$  NSs and to broaden their application in energy devices. Research has shown that assembling the ultrathin  $\text{NiCo}_2\text{O}_4$  NSs with graphene leads to an enhanced catalytic activity derived from the synergetic chemical coupling effects between ultrathin NSs and foreign materials [15,16].

Herein, we report the *in situ* growth of ultrafine  $\text{MnFe}_2\text{O}_4$  nanocrystals on ultrathin  $\text{NiCo}_2\text{O}_4$  NSs (denoted as  $\text{MnFe}_2\text{O}_4/\text{NiCo}_2\text{O}_4$ ) as a highly efficient and stable bifunctional electrocatalyst. Previous studies have shown that the spinel ferrite  $\text{MnFe}_2\text{O}_4$  is highly active towards ORR; research conducted by Sun et al demonstrates that the mono-dispersed  $\text{MnFe}_2\text{O}_4$  nanoparticles (NPs) supported on carbon support exhibits competitive ORR activity to the commercial Pt/C catalyst [17–19]. In an endeavor to boost the bifunctionality, sub-10 nm  $\text{MnFe}_2\text{O}_4$  nanocrystals have been grown on  $\text{NiCo}_2\text{O}_4$  ultrathin NSs. The ultrathin NS structure of  $\text{NiCo}_2\text{O}_4$  affords a high specific surface area and sufficient anchoring sites that allow subsequent integration of highly dispersed NPs. This structure enables an intimate contact between  $\text{MnFe}_2\text{O}_4$  NPs and  $\text{NiCo}_2\text{O}_4$  NSs, which is attributed to an efficient utilization of the catalyst surface and an extended electrochemically active surface area. The significantly enhanced bifunctional oxygen electrocatalytic activity is credited to the distinct architecture and complex composition of the  $\text{MnFe}_2\text{O}_4/\text{NiCo}_2\text{O}_4$  hybrid. The  $\text{MnFe}_2\text{O}_4/\text{NiCo}_2\text{O}_4$  hybrid electrocatalyst exhibits an overpotential of

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<https://doi.org/10.1016/j.apcatb.2018.05.047>

Received 22 February 2018; Received in revised form 14 May 2018; Accepted 15 May 2018

Available online 16 May 2018

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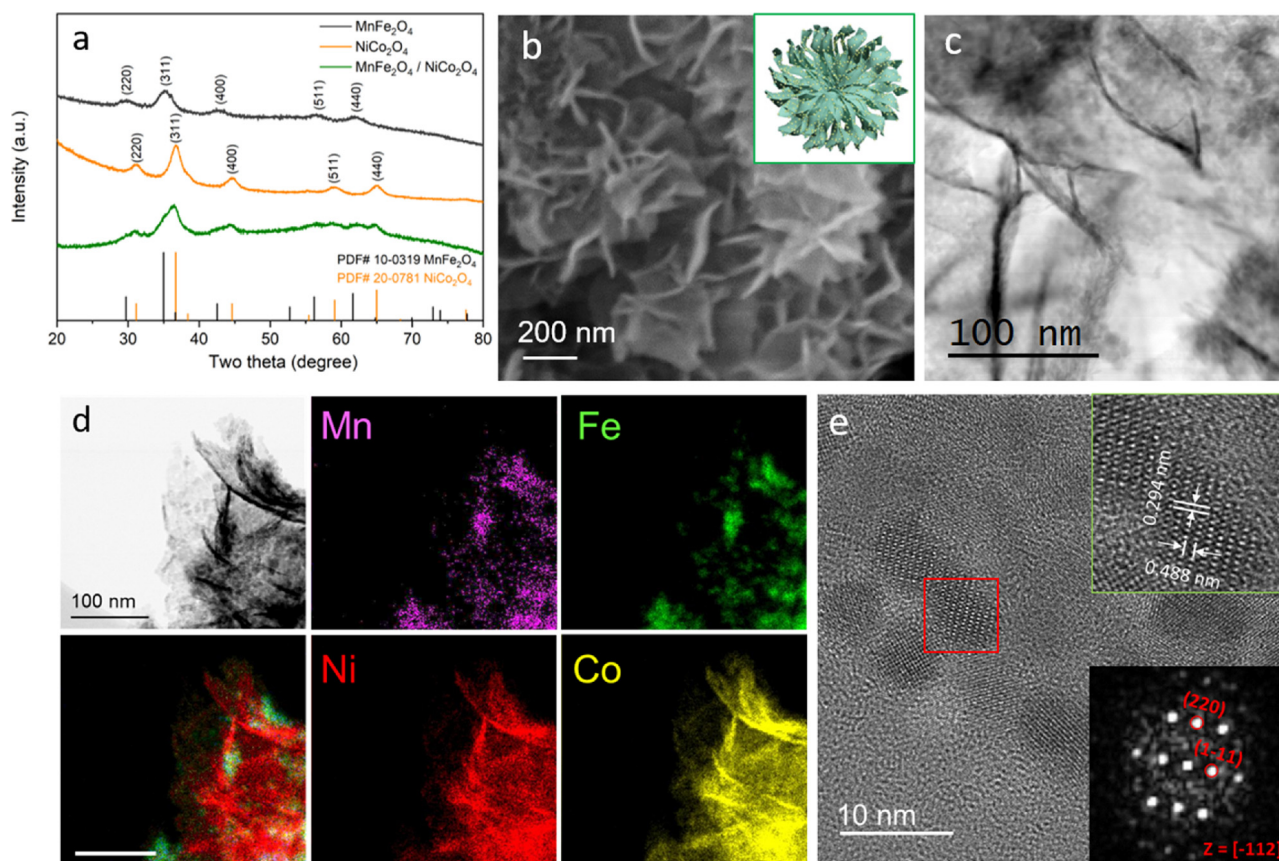
0.344 V at  $10 \text{ mA cm}^{-2}$  for the OER, and a positive half-wave potential of 0.767 V (vs. reversible hydrogen electrode, RHE) for the ORR. More importantly, the rechargeable Zn-air batteries employing the  $\text{MnFe}_2\text{O}_4/\text{NiCo}_2\text{O}_4$  hybrid achieve a high round-trip efficiency with a low discharge – charge voltage gap of 0.81 V at a reversible current density of  $10 \text{ mA cm}^{-2}$  as well as an excellent durability over 100 cycles.

## 2. Results and discussions

The ultrathin  $\text{NiCo}_2\text{O}_4$  NSs were firstly synthesized through a p-aminobenzoic acid (PABA) intercalation method, followed by a subsequent thermal treatment [7,8]. The as-synthesized  $\text{NiCo}_2\text{O}_4$  NSs exhibit negative zeta potentials in neutral and alkaline media (Fig. S1). The integration of  $\text{MnFe}_2\text{O}_4$  NPs on  $\text{NiCo}_2\text{O}_4$  NSs was achieved using a non-hydrolytic thermal decomposition method with  $\text{Mn}(\text{acac})_2$  and  $\text{Fe}(\text{acac})_3$  as the sources, and oleylamine as the solvent and stabilizer [20]. Subsequently, the negatively charged  $\text{NiCo}_2\text{O}_4$  NSs were introduced into the solution, which act as heterogeneous nucleation sites for the growth of  $\text{MnFe}_2\text{O}_4$  NPs, owing to the charge interaction of  $\text{Fe}^{3+}$  and  $\text{Mn}^{2+}$ . In this synthesis, the loading amount of  $\text{MnFe}_2\text{O}_4$  NPs was controlled by tailoring the ratio of  $\text{Fe}(\text{acac})_3$  and  $\text{Mn}(\text{acac})_2$  to the  $\text{NiCo}_2\text{O}_4$ . The strong anchoring of  $\text{MnFe}_2\text{O}_4$  resulted from the heterogeneous nucleation led to an interfacial interaction of  $\text{MnFe}_2\text{O}_4$  NPs with the ultrathin  $\text{NiCo}_2\text{O}_4$  NSs, which was expected to exhibit high electrocatalytic activity and durability for oxygen electrocatalysis. For comparison, monodispersed  $\text{MnFe}_2\text{O}_4$  NPs and physically mixed composite  $\text{MnFe}_2\text{O}_4 + \text{NiCo}_2\text{O}_4$  were also prepared.

The XRD pattern of the as-prepared electrocatalyst clearly demonstrates the formation of  $\text{MnFe}_2\text{O}_4/\text{NiCo}_2\text{O}_4$  hybrid (Fig. 1a), where the peak positions and relative intensities of all diffraction peaks are well matched with those of the resulting  $\text{MnFe}_2\text{O}_4$  NPs,  $\text{NiCo}_2\text{O}_4$  NSs and

corresponding standard diffraction patterns (PDF #10-0319 and PDF #20-0781). The FESEM image of  $\text{MnFe}_2\text{O}_4/\text{NiCo}_2\text{O}_4$  hybrid in Fig. 1b shows the layered flower-like structure assembled with interconnected ultrathin  $\text{NiCo}_2\text{O}_4$  NSs. Fig. 1c and d confirm the architecture of the  $\text{MnFe}_2\text{O}_4/\text{NiCo}_2\text{O}_4$  hybrid, where the  $\text{NiCo}_2\text{O}_4$  NSs are well integrated with  $\text{MnFe}_2\text{O}_4$  NPs. From the TEM images shown in Figs. S2 and S3, the thickness of an individual  $\text{NiCo}_2\text{O}_4$  NS matrix is around 6–7 nm. As revealed by the Brunauer–Emmett–Teller (BET) analysis (Fig. S4), the  $\text{MnFe}_2\text{O}_4/\text{NiCo}_2\text{O}_4$  hybrid possesses a specific surface area of  $118.6 \text{ m}^2 \text{ g}^{-1}$  with substantial mesopores (similar to that of  $\text{NiCo}_2\text{O}_4$  NSs, with  $131.8 \text{ m}^2 \text{ g}^{-1}$ ), suggesting negligible agglomeration of the hybrid after the incorporation of  $\text{MnFe}_2\text{O}_4$  NPs. This undoubtedly makes the  $\text{MnFe}_2\text{O}_4/\text{NiCo}_2\text{O}_4$  hybrid an excellent electrocatalyst with high surface area and abundant electroactive sites. The chemical compositions of the  $\text{MnFe}_2\text{O}_4/\text{NiCo}_2\text{O}_4$  hybrid,  $\text{NiCo}_2\text{O}_4$  ultrathin NS and  $\text{MnFe}_2\text{O}_4$  NPs were also confirmed by energy dispersive X-ray (EDX) spectroscopy (Figs. S5–S7) and inductively coupled plasma-atomic emission spectroscopy (ICP-AES); Based on ICP-AES result, the loading amount of  $\text{MnFe}_2\text{O}_4$  NPs was estimated to be 47.2 wt.% in the  $\text{MnFe}_2\text{O}_4/\text{NiCo}_2\text{O}_4$  hybrid, consistent with the set ratio. Moreover, the STEM-EDX and atom-scale high resolution TEM (HRTEM) in Fig. 1d and e give better insights into the chemical composition and material structure. The elemental mappings of  $\text{MnFe}_2\text{O}_4/\text{NiCo}_2\text{O}_4$  hybrid (Fig. 1d) show the homogenous distributions of Mn, Fe, Ni, Co, indicating that the ultrafine  $\text{MnFe}_2\text{O}_4$  NPs are uniformly dispersed on  $\text{NiCo}_2\text{O}_4$  NSs without deteriorating the structure and ultrathin feature of the  $\text{NiCo}_2\text{O}_4$  NSs during the formation of  $\text{MnFe}_2\text{O}_4$  NPs. As demonstrated, the exposed edges of  $\text{NiCo}_2\text{O}_4$  NSs are the electroactive centers for the water oxidation reaction and could remarkably facilitate the reaction kinetics [16]. To this end, the HRTEM was conducted; the lattice fringes with a d spacing of 2.51 Å (Fig. S3) on clean exposed edge



**Fig. 1.** (a) XRD of  $\text{MnFe}_2\text{O}_4$  NPs,  $\text{NiCo}_2\text{O}_4$  NSs and  $\text{MnFe}_2\text{O}_4/\text{NiCo}_2\text{O}_4$  hybrid; (b) FESEM image and (c) TEM image of  $\text{MnFe}_2\text{O}_4/\text{NiCo}_2\text{O}_4$  hybrid; (d) TEM image of  $\text{MnFe}_2\text{O}_4/\text{NiCo}_2\text{O}_4$  hybrid with EDX mapping; (The scale bar is 100 nm) (e) HRTEM images of  $\text{MnFe}_2\text{O}_4$  nanoparticles on  $\text{NiCo}_2\text{O}_4$  NSs.

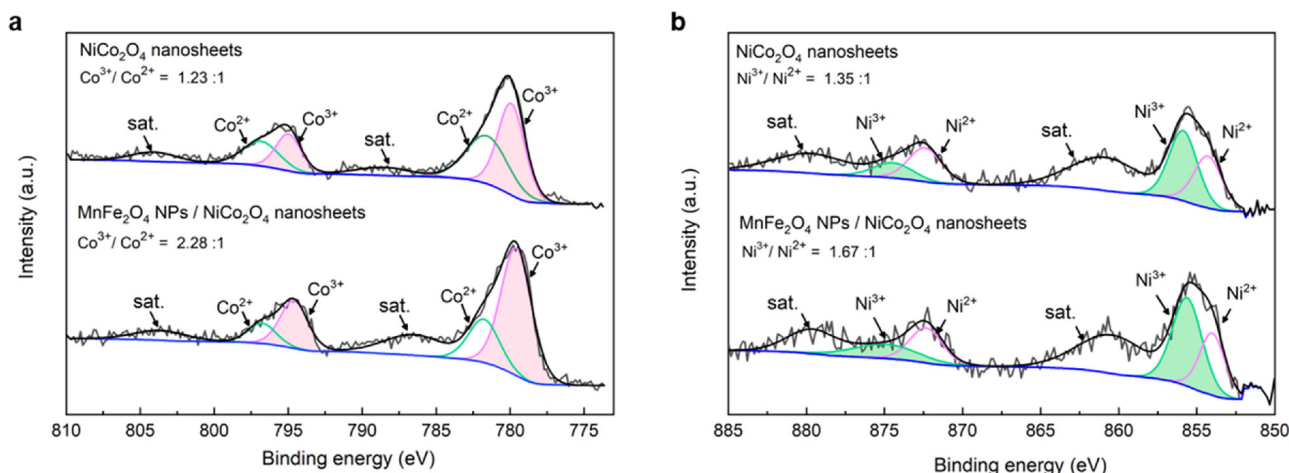


Fig. 2. The XPS profiles of NiCo<sub>2</sub>O<sub>4</sub> NSs and MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub> hybrid. (a) Co 2p spectra and its simulations; (b) Ni 2p spectra and its simulations.

of NiCo<sub>2</sub>O<sub>4</sub> NS were observed, which correspond to the (311) plane of NiCo<sub>2</sub>O<sub>4</sub>, implying that the NiCo<sub>2</sub>O<sub>4</sub> is preferentially exposed with {311} facets [9]. As seen from Fig. 1e, the diameters of the NPs coupled on the NiCo<sub>2</sub>O<sub>4</sub> are 5–7 nm. The HRTEM of MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub> hybrid, along with the corresponding diffractogram, shows the d spacings of 2.94 and 4.88 Å, indexed on the (220) and (1-11) planes of MnFe<sub>2</sub>O<sub>4</sub>. This HRTEM confirms that the non-hydrolytic synthetic method offers intriguing advantages to synthesize NPs with uniform size distribution and high crystallinity [20,21].

X-ray photoelectron spectroscopy (XPS) analysis was performed to study the electronic structure of Ni, Co, O in the MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub> hybrid and the ultrathin NiCo<sub>2</sub>O<sub>4</sub> NSs. After loading with MnFe<sub>2</sub>O<sub>4</sub> NPs, the Co 2p core level XPS spectra demonstrate that the electron binding energy of Co 2p experiences a decrease of ~0.4 eV. This is considered to accompany the oxidation of Co<sup>2+</sup> to Co<sup>3+</sup> [22]. The Co 2p core level spectra (Fig. 2a) were deconvoluted into two spin-orbit doublets, characteristic of Co<sup>2+</sup> and Co<sup>3+</sup>, and one shakeup satellite (identified as “Sat.”). The deconvolution results manifest that, after loading with MnFe<sub>2</sub>O<sub>4</sub> NPs, the ratio of Co<sup>3+</sup> to Co<sup>2+</sup> experienced a significant increase. It has been reported that Co<sup>3+</sup> center plays a decisive role in determining H<sub>2</sub>O adsorption, and consequently enabling high ORR and OER electrocatalytic activities [19,23]. As for the Ni 2p spectra (Fig. 2b), two spin orbit doublets, characteristic of Ni<sup>2+</sup> and Ni<sup>3+</sup> species, are found in both samples. The incorporated surface Ni<sup>3+</sup> species also acts as the active sites to facilitate the OER [24].

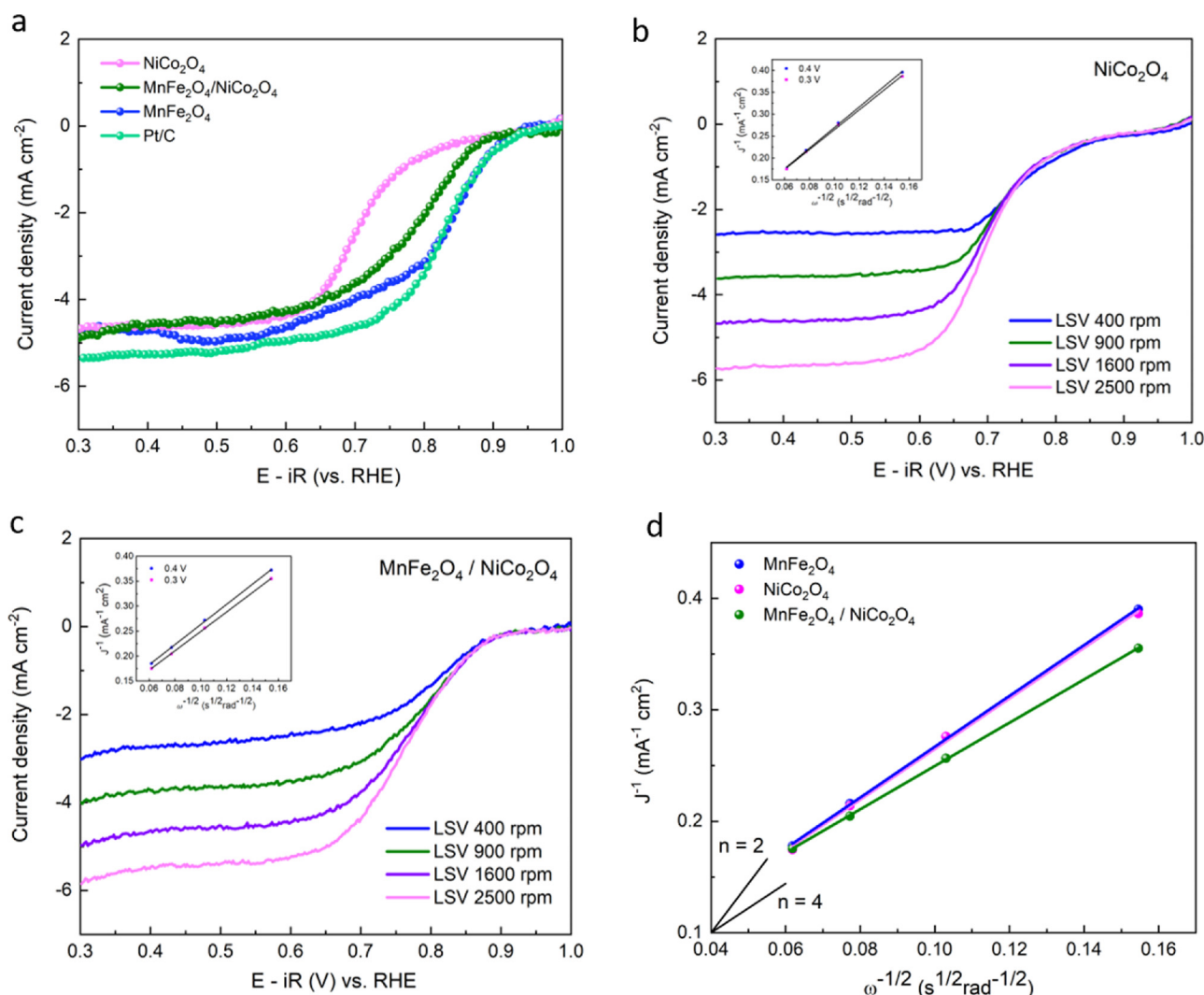
Research has shown that the complex transition metal oxide generally results in complementary redox and beneficial electronic properties, and therefore is likely to show significant promise as electrocatalyst material [25]. As such, we investigated the bifunctional oxygen electrocatalytic performances of the MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub> hybrid for ORR and OER. The ORR electrocatalytic activities were first investigated by the cyclic voltammetry (CV) in O<sub>2</sub> and Ar saturated 0.1 M KOH, respectively. Using the MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub> hybrid, a more positive onset potential was achieved (Fig. S12), and the peak potential shifted from 0.695 V (NiCo<sub>2</sub>O<sub>4</sub> NSs) to 0.770 V vs. Reversible Hydrogen Electrode (RHE). To reveal the ORR kinetics, the catalytic activity of the MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub> hybrid was further analyzed by the linear sweep voltammetry (LSV), using a rotating disk electrode (RDE) with different rotation speeds in O<sub>2</sub>-saturated 0.1 M KOH. As benchmarks, the ORR catalytic activities of MnFe<sub>2</sub>O<sub>4</sub> NPs, NiCo<sub>2</sub>O<sub>4</sub> NSs and physically mixed composite as well as the commercial Pt/C electrocatalysts were evaluated. At 1600 rpm (Fig. 3a), the NiCo<sub>2</sub>O<sub>4</sub> NSs showed a diffusion-limited current density of 4.66 mA cm<sup>-2</sup> at 0.3 V (vs. RHE). The ORR E<sub>onset</sub> and E<sub>1/2</sub> of the NiCo<sub>2</sub>O<sub>4</sub> were 0.785 and 0.702 V, respectively. When coupled with MnFe<sub>2</sub>O<sub>4</sub>, the MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub> hybrid achieved a comparably higher catalytic ORR activity, with a more positive onset

potential (E<sub>onset</sub> 0.881 V vs. RHE), half wave potential (E<sub>1/2</sub> 0.767 V), and higher diffusion-limited current density (5.01 mA cm<sup>-2</sup> at 0.3 V). The physically mixed composite demonstrated a similar ORR electrocatalytic activity (Fig. S14), with an onset potential of 0.878 V and a half-wave potential of 0.783 V. It is suggested that the redox couple of Mn<sup>3+</sup>/Mn<sup>2+</sup> within the hybrid attributes to an enhanced ORR activity. As Mn<sup>2+</sup> has a higher tendency to adsorb O<sub>2</sub>, the surface redox active centers of MnFe<sub>2</sub>O<sub>4</sub> presumably facilitate the O<sub>2</sub> adsorption and activation in the ORR [19,26]. With the onset potential of 0.912 V and half-wave potential of 0.835 V, MnFe<sub>2</sub>O<sub>4</sub> NPs demonstrate the best ORR activity among the resulting spinel oxide catalysts, approaching to the commercial Pt/C catalyst performance wise. To further investigate the reaction kinetics of the hybrid for ORR, polarization curves were obtained on all samples at the rotating speeds of 400 to 2500 rpm (Fig. 3b and c, Figs. S10 and S11). The electron transfer numbers were estimated based on the slopes Koutecky-Levich equation at different potentials (ie., 0.3 V and 0.4 V). The electron transfer numbers (n) were calculated to be around 4.0 at 0.3–0.4 V for MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub> hybrid, slightly higher than that of NiCo<sub>2</sub>O<sub>4</sub> NSs. This indicates that the electron transfer pathway is dominated by a four-electron route in the MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub> hybrid.

The LSV curves of the MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub> electrocatalyst for OER were measured at 1600 rpm in O<sub>2</sub> saturated 0.1 M KOH (Fig. 4a). Remarkably, the MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub> achieved the lowest overpotential of 344 mV at the current density of 10 mA cm<sup>-2</sup>, significantly lower than that of NiCo<sub>2</sub>O<sub>4</sub> NSs (456 mV). Of note, the MnFe<sub>2</sub>O<sub>4</sub> NPs possess poor OER activity. Physically mixed composite, MnFe<sub>2</sub>O<sub>4</sub> + NiCo<sub>2</sub>O<sub>4</sub>, also showed an improved OER catalytic activity (418 mV) as compared to the NiCo<sub>2</sub>O<sub>4</sub> NSs. The remarkable OER activity of the MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub> hybrid mainly originated from the NiCo<sub>2</sub>O<sub>4</sub> NSs, where MnFe<sub>2</sub>O<sub>4</sub> served as a synergist. It is also worth noting that the MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub> hybrid outperforms the MnFe<sub>2</sub>O<sub>4</sub> + NiCo<sub>2</sub>O<sub>4</sub> composite, with the overpotential (at 10 mA cm<sup>-2</sup>) being 74 mV lower than the MnFe<sub>2</sub>O<sub>4</sub> + NiCo<sub>2</sub>O<sub>4</sub> electrocatalyst (Fig. S15). It is deduced that the strong interfacial effects between MnFe<sub>2</sub>O<sub>4</sub> NPs and NiCo<sub>2</sub>O<sub>4</sub> NSs in the MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub> will contribute to a more efficient charge transfer between the NPs and the support NSs in comparison to the physically mixed composite [27]. The superb OER activity is also ascribed to the dual electroactive sites, i.e., the clean edge of the NiCo<sub>2</sub>O<sub>4</sub> ultrathin NSs and the interfacial coupling sites between MnFe<sub>2</sub>O<sub>4</sub> NPs and NiCo<sub>2</sub>O<sub>4</sub> NSs.

Importantly, the MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub> hybrid demonstrated a smaller Tafel slope (46.7 mV dec<sup>-1</sup>) than NiCo<sub>2</sub>O<sub>4</sub> NSs (77.9 mV dec<sup>-1</sup>) and MnFe<sub>2</sub>O<sub>4</sub> NPs (115.5 mV dec<sup>-1</sup>), as shown in Fig. 4b. A lower Tafel slope manifests a better OER activity as the electrode experiences smaller polarization upon increasing anodic current density [28].





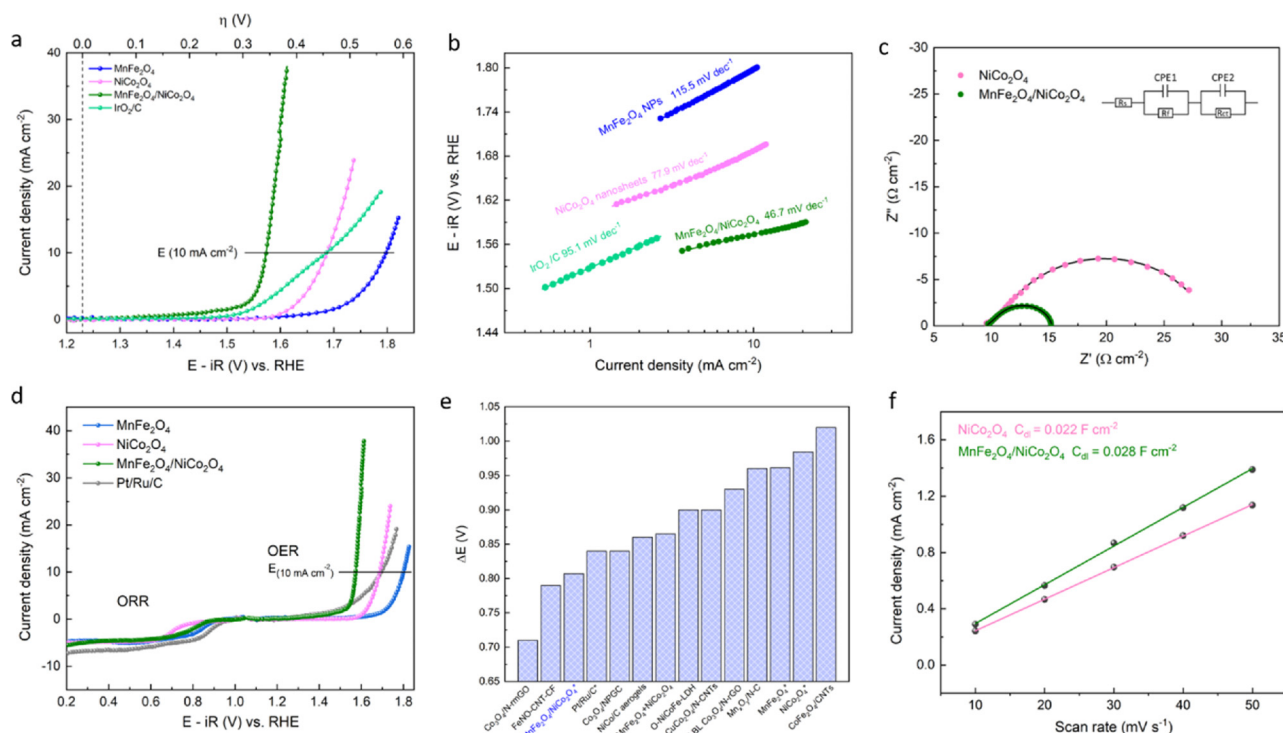
**Fig. 3.** (a) ORR LSV curves at 1600 rpm for NiCo<sub>2</sub>O<sub>4</sub> NSs, MnFe<sub>2</sub>O<sub>4</sub> NPs, MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub>, and Pt/C. ORR LSV curves of (b) NiCo<sub>2</sub>O<sub>4</sub> NSs and (c) MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub> hybrid in 0.1 M KOH at different rotation speeds (400, 900, 1600, and 2500 rpm). (d) Electron transfer number of NiCo<sub>2</sub>O<sub>4</sub> NSs, MnFe<sub>2</sub>O<sub>4</sub> NPs, MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub> derived from K–L plots at 0.3 V vs RHE.

Potentiostatic electrochemical impedance spectroscopy (EIS) conducted on NiCo<sub>2</sub>O<sub>4</sub> ultrathin NSs and MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub> hybrid (Fig. 4c) further verifies the enhanced interfacial properties of the MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub> hybrid under the OER potential. For example, at a bias potential of 0.65 vs. Ag/AgCl, the MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub> hybrid shows an interfacial charge-transfer resistance of 3.73  $\Omega$  cm<sup>2</sup> (Table S1), dramatically lower than that of NiCo<sub>2</sub>O<sub>4</sub> (15.68  $\Omega$  cm<sup>2</sup>, Table S1). The simulated results indicate that the MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub> hybrid possesses a faster interfacial charge transfer process. A further enhanced charge-transfer process was found on the MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub> hybrid when 1 M KOH was used as the electrolyte (Fig. S17). Apart from the substantially enhanced electrocatalytic activity, the stability of the MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub> hybrid is also of great significance for the practical application [29]. To assess the stability, the galvanostatic long-term performances of MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub> were conducted at a constant current density load of 10 mA cm<sup>-2</sup> in O<sub>2</sub>-saturated 1 M KOH at 1600 rpm, as shown in Fig. S18. The OER potential of the MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub> hybrid was first plateaued at 1.56 V at a current density of 10 mA cm<sup>-2</sup>, and the overpotential only experienced a 9 mV increase after polarized for 50 000 s.

The overall bifunctional activity of the MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub> hybrid

was further confirmed by the potential difference between the OER potential of E<sub>10</sub> and the ORR half-wave potential ( $\Delta E = E_{10} - E_{1/2}$ ) in 0.1 M KOH. The bifunctional activities of the MnFe<sub>2</sub>O<sub>4</sub> NPs, NiCo<sub>2</sub>O<sub>4</sub> ultrathin NSs and Pt/Ru/C were also evaluated as shown in Fig. 4d. The MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub> hybrid shows the smallest potential difference (0.807 V) among the catalysts studied, and is comparable with many good bifunctional catalysts in alkaline conditions (Fig. 4e, listed in Table S2), which is indicative of the favorable bifunctionality of MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub>. By integrating the MnFe<sub>2</sub>O<sub>4</sub> NPs on the ultrathin NiCo<sub>2</sub>O<sub>4</sub> NSs, the overall overpotential was successfully decreased by 177 mV.

On account of the architecture, chemical compositions and valence states of the MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub> hybrid, it is believed that the advanced bifunctionality is mainly attributed to the following two foundations. The uniform distribution of the ultrafine MnFe<sub>2</sub>O<sub>4</sub> NPs on the porous ultrathin NiCo<sub>2</sub>O<sub>4</sub> NSs results in a high electrolyte – material contact area and short ion diffusion paths, which facilitates the H<sub>2</sub>O molecule adsorption and favors the charge-transfer [30,31]; the strong coupling of the MnFe<sub>2</sub>O<sub>4</sub> NPs on the NiCo<sub>2</sub>O<sub>4</sub> NSs suppresses the agglomeration of the nanoparticles during the long-term operation [27,32]. In



**Fig. 4.** LSV curves of (a) OER reaction on NiCo<sub>2</sub>O<sub>4</sub> NSs, MnFe<sub>2</sub>O<sub>4</sub> NPs, MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub>, and IrO<sub>2</sub>/C in 0.1 M KOH at 1600 rpm. (b) Tafel curves of on NiCo<sub>2</sub>O<sub>4</sub> NSs, MnFe<sub>2</sub>O<sub>4</sub> NPs, MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub>, and IrO<sub>2</sub>/C. (c) EIS curves of MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub> hybrid and NiCo<sub>2</sub>O<sub>4</sub> NSs in 0.1 M KOH at a bias potential of 0.65 V vs. Ag/AgCl. Inset: equivalent circuit model for the Nyquist plots of the NiCo<sub>2</sub>O<sub>4</sub> NSs and the MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub> hybrid electrocatalysts. (The  $R_s$ ,  $R_f$  and  $R_{ct}$  represent the electrolyte, electrode film and charge transfer resistance, respectively. CPE1, CPE2 refer to the constant phase elements of the electrode film, and charge-transfer double-layer components.) (d) LSV curves of an overall oxygen reaction on MnFe<sub>2</sub>O<sub>4</sub> NPs, NiCo<sub>2</sub>O<sub>4</sub> NSs, MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub> hybrid and Pt/Ru/C. (e) The comparison of and overall overpotential for bifunctional electrocatalysts. (f) Measurements of the electrochemical double-layer capacitance of the NiCo<sub>2</sub>O<sub>4</sub> NSs and the MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub> hybrid at current density at the potential of 1.275 V vs. scan rate.

addition, a larger electrochemical double layer capacitance was found on MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub>, suggesting an extended electrochemical active surface area [33]. The capacitance of the double layer ( $C_{dl}$ ) was evaluated by measuring the CV curves at a non-faradic potential range at different scan rates (Fig. 4f and S19) [34,35]; the  $C_{dl}$  of MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub> is 0.028 F cm<sup>-2</sup>, larger than that of NiCo<sub>2</sub>O<sub>4</sub> (0.022 F cm<sup>-2</sup>). This is likely to be one of the reasons for the enhanced electrocatalytic activity from the MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub> hybrid. This heterostructured hybrid realizes efficient bifunctional catalytic activities from the full utilization of the catalyst surface and strong interaction between the MnFe<sub>2</sub>O<sub>4</sub> NPs and NiCo<sub>2</sub>O<sub>4</sub> matrix. The decreased overall potential difference between ORR and OER on MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub> indicates a small energy loss in the charging/discharging process.

Building on bifunctional electrochemical activities studied above on the RDE electrode, the oxygen catalytic activity of the MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub> hybrid was further evaluated in the rechargeable Zn-air battery. As shown in Fig. 5a (see the photo in Fig. S20), a proof-of-concept rechargeable Zn-air battery was assembled with zinc plate as the anode, oxygen electrocatalyst coated gas diffusion layer (GDL) as the cathode. Fig. 5b shows typical discharge – charge polarization curves of the rechargeable Zn-air battery. The MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub> hybrid exhibits a discharge – charge voltage gap of 0.80 V at 10 mA cm<sup>-2</sup> and 1.10 V at 50 mA cm<sup>-2</sup>, which are significant lower than those of NiCo<sub>2</sub>O<sub>4</sub> (0.94 V at 10 mA cm<sup>-2</sup> and 1.28 V at 50 mA cm<sup>-2</sup>) and commercial Pt/Ru/C catalyst (0.87 at 10 mA cm<sup>-2</sup> and 1.25 V at 50 mA cm<sup>-2</sup>). The small voltage gaps between charging and discharging processes on MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub> hybrid suggest an enhanced voltaic efficiency of the MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub> hybrid. More importantly, the hybrid catalyst exhibits

excellent durability with stable efficiency in Fig. 5c. When being repeatedly charged-discharged at a constant current density of 10 mA cm<sup>-2</sup> (10 min charging, 10 min discharging), no observable degradation was found on the MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub> hybrid after 100 cycles. The voltaic efficiency was found to be 59.7% at the beginning and experienced a slight increase after 32 h of the operation (60.8%). In contrast, the Pt/Ru/C shows a much inferior stability during the charging-discharging process with the voltaic efficiency decreasing from 56.4% to 49.7%, caused by the passivation of Pt at positive potentials pertinent to OER and electrocatalyst detachment [4,36].

### 3. Conclusions

In summary, we reported a desirable approach to preparing an efficient bifunctional oxygen electrocatalyst through synergistically integrating non-precious metal oxide nanoparticles on ultrathin NSs. This inexpensive, earth-abundant hybrid catalyst exhibits superior bifunctional catalytic activity in terms of its lower overall overpotential ( $\Delta E = 0.807$  V) than the MnFe<sub>2</sub>O<sub>4</sub> and NiCo<sub>2</sub>O<sub>4</sub> electrocatalyst. The fact that the MnFe<sub>2</sub>O<sub>4</sub>/NiCo<sub>2</sub>O<sub>4</sub> hybrid outperforms the state-of-the-art precious metals Pt/Ru/C with respect to bifunctional activity and durability makes the discovery in this study more promising. We believe that the large surface area, high chemical stability of NiCo<sub>2</sub>O<sub>4</sub> NSs and strong chemical coupling effects between MnFe<sub>2</sub>O<sub>4</sub> NPs and NiCo<sub>2</sub>O<sub>4</sub> afford the enhanced bifunctional catalytic activity and durability. This work holds the promise to open a new possibility in designing novel transition metal based bifunctional catalysts as the alternatives to the noble metals for the application in energy related devices.

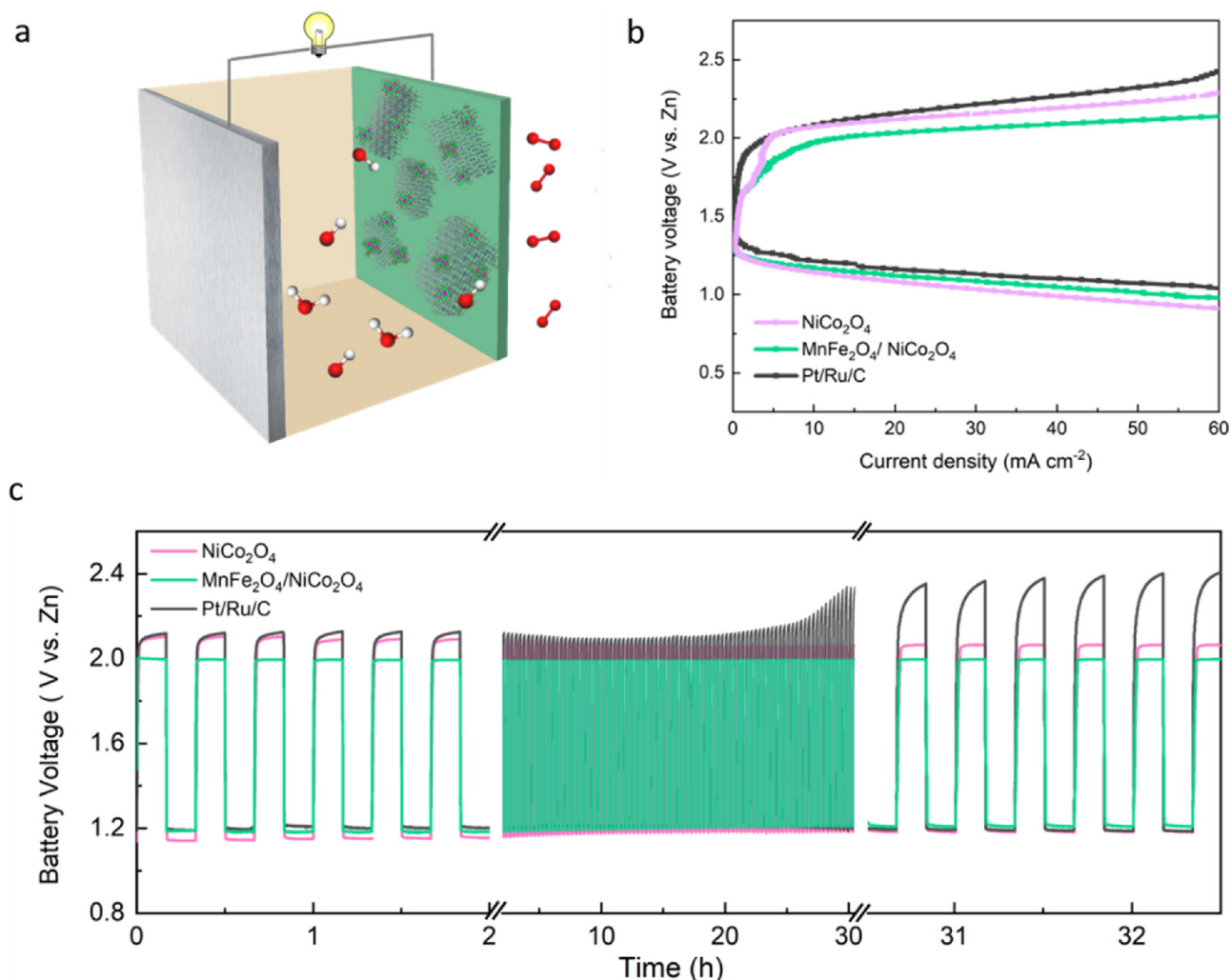


Fig. 5. (a) The schematic figure of the custom built rechargeable Zn – air batteries. (b) Charge and discharge polarization curves of  $\text{NiCo}_2\text{O}_4$  NSs,  $\text{MnFe}_2\text{O}_4/\text{NiCo}_2\text{O}_4$ , and Pt/Ru/C air electrode. (c) Galvanostatic charge – discharge curves at  $10 \text{ mA cm}^{-2}$  using  $\text{NiCo}_2\text{O}_4$  NSs,  $\text{MnFe}_2\text{O}_4/\text{NiCo}_2\text{O}_4$ , and Pt/Ru/C air electrode.

## Acknowledgements

This work was financially supported by the Natural Sciences and Engineering Research Council of Canada (NSERC), Discovery Grant under contract of RGPIN-2016-05494 and Alberta Innovates Technology Futures (AITF) Graduate Student Scholarship.

## Appendix A. Supplementary data

Supplementary material related to this article can be found, in the online version, at doi: <https://doi.org/10.1016/j.apcatb.2018.05.047>.

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